

Research Article

Preparation of Aluminium-Silver Composite Cutting Tool by Chemical Synthesis

Jayashree Das^{Å*}, Prakash Kumar Sahu^B, Biswajit Parida^B, Ashim Guha^C and Punya Priya Mishra^Å

^AMechanical Engineering Department, Veer Surendra Sai University of Technology, Burla, Orissa, India ^BMechanical Engineering Department, Indian Institute of Technology, Guwahati, Assam, India ^CMechanical Engineering Department, Bengal Engineering and Science University, Shibpur, West Bengal, India

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Abstract

The present work is aimed to develop metal toughened aluminium-silver composite which will be used as a cutting tool material. Chemical synthesis using the sol-gel method was carried-out to prepare alumina powder from aluminium nitrate $Al (NO_3)_3, 9H_2O$. Then $AgNO_3$ was well mixed with the prepared α - Al_2O_3 powder in aqueous solution followed by evaporation and heating, to get 10 volume percent of Ag in the Al_2O_3 -Ag mixture. A conventional powder metallurgical route was followed to prepare Al_2O_3 -Ag composite pellets. The pellets were first dried at 500°C for 2 hours to burn off the PVA, then pre-sintered at 950°C for 2 hours and finally sintered at 1600°C for 2 hours as soaking time in a furnace. XRD test shows that the phase present in the powder heated at 1600°C for 2 hours is α -alumina and only silver is present in the developed pellet as a second phase. Bulk density and open porosity of the samples were measured using Archimedes principle and employing boiling water method. Bulk densities of some of the composite pellets were found nearly 3 g/cc. Attempt should be made to improve the density in order to use the developed powder as cutting tool material. Ceramic cutting tools can be used in metal cutting, and can keep their hardness, strength, abrasion resistance and long performance life under high cutting speed. With these special mechanical properties, ceramic cutting tools will certainly substitute high speed steel cutting tools in many fields.

Keywords: Aluminium-silver composite, Sol-gel method, Bulk-density, Open porosity, Hardness

1. Introduction

1.1 Cutting Tool

The demand of the manufacturing industries in the latter half of nineteenth century has led to the development of improved machine tools, cutting tools and production processes. The cutting tool is one of the important elements in realizing the full potential out of any metal cutting operation. Tool material have improved rapidly during the last few decades, and in many instances the development of new tool material has necessitated change in the design trend of machining tool to make full use of the potentialities of tool materials for higher productivities. Progress from carbon tool steel, HSS and cast alloys to carbide and ceramics has facilitated the application of higher speed at each stage of development, because improvements achieved in properties of tool materials. Among the available tool materials we have chosen ceramic for further development of cutting tool.

1.2 Ceramic Cutting Tool

Ceramics are nonmetallic materials. This puts them in an entirely different category than HSS and carbide tool

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materials. The use of ceramics as cutting tool material has distinct advantages and disadvantages. The application of ceramic cutting tools is limited because of their extreme brittleness. The transverse rupture strength (TRS) is very low. This means that they will fracture more easily when making heavy or interrupted cuts. However, the strength of ceramics under compression is much higher than HSS and carbide tools. There are two basic types of ceramic material: hot-pressed and cold-pressed. In hot-pressed ceramics, usually black or gray in color, the aluminum oxide grains are pressed together under extremely high pressure and at a very high temperature to form a billet. The billet is then cut to insert size. With cold-pressed ceramics, usually white in color, the aluminum oxide grains are pressed together, again under extremely high pressure but at a lower temperature.

Ceramics may not be the all-around tool for the average shape they can be useful in certain applications. Ceramic tools have been alloyed with zirconium (about 15%) to increase their strength. Many ceramic tool manufacturers are recommending the use of ceramic tools for both rough cutting and finishing operations. Practical shop experience indicates that these recommendations are somewhat optimistic. To use ceramic tools successfully, insert shape, work material condition, machine tool capability, set-up, and general machining conditions must all be correct. High rigidity of machine tool and set-up is Dearnley et al. (1982) studied the stringent and complex demands put upon tool materials, it is often said that the ideal material should therefore have the hardness of diamond, the toughness of high-speed steel and the chemical inertness of alumina. A. Senthil Kumar et al. (2007) developed, yttria and ceria toughened alumina (YCTA) ternary ceramic composite for cutting tool application. Ersan Aslan et al. (2007) present the paper outlines an experimental study to achieve this by employing Taguchi techniques. Combined effects of three cutting parameters, namely cutting speed, feed rate and depth of cut on two performance measures, flank wear and surface roughness (R_a) , were investigated employing an orthogonal array and the analysis of variance (ANOVA). Shi-Xue Song et al. (2003) was fabricated Al₂O₃/Ti (C_{0.3}N_{0.7}) cutting tool material successfully using a new kind of sintering technique, i.e. repetitious-hot-pressing technique. Yang et al. (1969) investigated the strengthening and toughening mechanisms of hot-pressed Ce-TZP/Al₂O₃ ceramics. B. Smuk et al. (2003) presented a series of ceramic tool materials based on Al₂O₃ with ZrO₂. Experiments on ceramic materials with different contents of ZrO₂ addition and different phase compositions in the delivered state were carried out. Huang et al. (2002) fabricated Al₂O₃/ (W, Ti) C composite ceramic materials reinforced by rare earth additives by the hot pressing technique. Microstructure, flexural strength, toughening and strengthening mechanisms are analyzed. The maximum flexural strength and fracture toughness of the yttrium reinforced Al₂O₃/ (W, Ti) C ceramic are 853 MPa and 6.1 MPa, respectively, which are approximately 20% and 16% higher than for the corresponding material without yttrium. Chonghai Xu et al. (2006) studied that the addition of rare earth additives such as yttrium of a suitable amount in a proper way can notably improve the flexural strength and fracture toughness and fracture resistance of Al₂O₃ /(W, Ti)C ceramic material. L.A. Dobrza nski et al. (2005) presented investigation results of structure and properties of the coatings deposited with the PVD and CVD techniques on cutting inserts made from the $Al_2O_3 + ZrO_2$ oxide tool ceramics.

2. Experimental Work

2.1 Chemical Synthesis

In a typical experiment, boehmite solution was prepared by hydrolyzing aluminium nitrate, Al $(NO_3)_3,9H_2O$ dissolved in distilled water followed by peptization using nitric acid. The obtained boehmite was first heated in a sand bath heater, and then heating was done in a medium temperature furnace and then high temperature furnace in subsequent temperatures to get α -alumina. Some amount of prepared powder (~2g) was separated for X-ray diffraction.

2.2 Preparation of Alumina Powder by Chemical Synthesis

i. 56.752 gm aluminium nitrate (Al $(NO_3)_3$, 9H₂O, AR Grade MERCK Ltd., India) was taken in a large beaker

(2000ml) and dissolved in 400ml distilled water (Figure-2.1[a]).

ii. Then the beaker was placed on a magnetic stirrer- cum heater and kept under stirring. The temperature was maintained at ~363K, (Figure- 2.1 [b]).

iii. Ammonium hydroxide (NH₄OH, *i.e.* 25% NH₃ in water) was slowly added to the solution with the help of a glass dropper, (Figure- 2.1[c]).

iv. Gel formation was observed. Controlled addition of NH₄OH was carried out till gel formation continued.

v. The gel was then filtered (~41 hours), (Figure- 2.2[d, e]).

vi. The filtered gel (boehmite, AlOOH) was taken out and kept in a separate beaker. Some amount of distilled water (~400ml) was added, (Figure- 2.3[f]).

vii. It was very slowly (drop by drop) peptized by 1:1 solution by distilled water (20ml) and HNO₃ (20ml), (GR 70% MERCK Ltd., India) keeping the temperature ~333K. The peptization process was continued till pH 3.5 of the solution was achieved, Figure- 2.3[g]).

viii. The mixture was kept under magnetic stirring for a period of 4 hours, (Figure- 2.3[h]).

ix. The solution was heated at ~ 493K for nearly 13 hours in a sand-bath heater,(Figure-2.4[i, j, k]).

x. The obtained powder was ground in an Agate pestle and mortar and weight was taken, (Figure- 2.5[1, m]).

xi. The ground powder was first heated at 500°C for 2 hours in a furnace, furnace cooled and powder weight was taken after cooling 6.082 gm then obtained powder was ground in an Agate pestle and mortar, weight was taken 6.066 gm. Then heated at 1200°C for 2 hours (Figure-2.6[n, o]), furnace cooled, powder Weight was taken after cooling 5.584 gm and again the obtained powder was ground in a Agate pestle and mortar , finally powder weight was taken 5.562gm, (Figure-2.7).

2.3 Flow Chart for Preparation of α -Alumina Powder



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2.4 Illustration of the Project Work is Shown Bellow



Figure 2.7- Alpha Alumina Powder

2.5 Flow Chart to Prepare Alumina-Silver Composite Pellet



2.6 Prepared Alumina-Silver Composite Pellet



Al₂O₃ Pellets, Sample-1, 2 & 3 Sample-4 & 5

Al₂O₃+Ag Pellets,

3. Result and Discussion

Alumina-silver composite has been successfully developed by solution sol-gel route and conventional powder metallurgical process. Test of the nature of the composites and characterization of their properties are necessary supplements. These aspects are presented in the different subsections.

3.1 Phase Identification

Phase identification of the developed powder heated at 500°C, 950°C and 1600 °C respectively for 2 hours was made by X-ray diffraction analysis using an X-ray diffract meter.



Fig. 3.1 XRD scan for the developed powder heated at 500°C for 2 hours. Phase: boehmite– AlO (OH)



Fig. 3.2 XRD scan for the developed powder heated at 950°C for 2 hours. Phases: boehmite– AlO (OH), α -alumina – Al₂O₃

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Fig. 3.4 XRD scan for the developed pellet sintered at 950° C for 2 hours. Phases: α -alumina – Al₂O₃, Silver – Ag



Fig.3.5 XRD scan for the developed pellet sintered at 1600°C for 2 hours. Phases: α -alumina – Al₂O₃, Silver–Ag

From the XRD analysis, fig 3.1 shows that the phase present in the powder heated at 500°C for 2 hours is boehmite AlO(OH) (Ref. 083-2384 JCPDS). Fig 3.2 shows for the powder heated at 950°C for 2 hours, phases present are boehmite AlO(OH), α -alumina – Al₂O₃(821-468 JCPDS). Fig 3.3 shows for the powder heated at

1600°C for 2 hours, phase present is α -alumina – Al₂O₃. Fig 3.4 shows that phases present in the pre-sintered pellet at 950°C for 2 hours are α -alumina – Al₂O₃, Silver – Ag (Ref.04-0783 JCPDS). Fig 3.5 also shows that phases present in the sintered pellet at 1600°C for 2 hours are α alumina – Al₂O₃, Silver – Ag.

3.2 Measurement of Sintered Density

Table3.1 Green densities of pellet

Material	Diameter of pellet (\u00f6) mm	Thickness of pellet (L) mm	Volume (V) cc	Weight (W) g	Green density g/cc
Al ₂ O ₃	10.34	7.62	0.637	1.258	1.9748
	10.34	4.42	0.371	0.7494	2.0199
	10.34	2.50	0.210	0.4245	2.0223
Al ₂ O ₃	10.34	4.82	0.404	0.8315	2.0546
+Ag	10.34	2.56	0.215	0.4909	2.2843

Table3.2 Densities of pellets heated after 500° C

Material	Diameter of pellet (\$) mm	Thickness of pellet (L) mm	Volume (V) cc	Weight (W) g	Density g/cc
Al ₂ O ₃	10.34	7.62	0.63818	1.2287	1.9253
	10.34	4.56	0.38290	0.7288	1.9033
	10.34	2.62	0.22004	0.4112	1.8690
Al ₂ O ₃	10.34	4.82	0.39485	0.7796	1.9739
+Ag	10.34	2.56	0.21830	0.4621	2.1167

Table3.3 Pre-sintered densities of pellets heated at 950[°] C

Material	Diameter of pellet	Thickness of pellet	Volume (V) cc	Weight (W) g	Density g/cc
	(ø) mm	(L) mm			•
	10.32	7.62	0.6373	1.2271	1.9252
Al_2O_3	10.32	4.52	0.3780	0.7269	1.9226
	10.32	2.68	0.2241	0.4097	1.8276
Al ₂ O ₃	10.36	4.74	0.3995	0.7787	1.9488
+Ag	10.32	2.72	0.2275	0.4610	2.0260

Table3.4 Sintered densities of pellets heated at1600°C

Material	Diameter of pellet (φ) mm	Thickness of pellet (L) mm	Volume (V) cc	Weight (W) g	density g/cc
Al ₂ O ₃	9.16	6.72	0.44284	1.2219	2.7590
	9.22	4.05	0.26706	0.7223	2.7048
	9.22	2.36	0.15688	0.4080	2.6006
Al ₂ O ₃	9.00	4.02	0.25574	0.7215	2.8212
+Ag	9.02	2.12	0.13546	0.4065	3.0009

Table 3.1 shows the green densities of the pellets compacted at 250MPa pressure. Densities of the pellets heated at 500°C, pre-sintered densities at 950°C and sintered densities at 1600°C are shown in table 3.2, 3.3 and 3.4 respectively. The sample pellets are first heated at 500°C to burn off the polyvinyl alcohol (PVA) binder. Due to the above mentioned weight loss; densities of the pellets have decreased a little. The pellets were then presintered at 950°C for 2 hours. At this stage the weight loss

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is mainly due to evaporation of silver from the outer surface of the pellets. The pellets were finally sintered at 1600°C for 2 hours. Sintered densities of the developed alumina-silver composite pellets ware found higher than that of pure alumina pellet.

3.3 Bulk density and open porosity of the samples

Table3.5 Bulk density and open porosity of the samples

Material	Dry	Suspende	Saturated	Bulk	Open
	weight	d weight	weight	density	porosit
	(W _d)g	(W _{sus}) g	(W _{sat})g	$(\rho)g/cc$	v
		(545) 0	(Jac, D	0.20	(p _o)
	1.240	0.350	1.281	3.54	11.70
Al_2O_3	0.732	0.230	0.761	3.18	12.60
	0.413	0.111	0.425	3.72	10.80
Al ₂ O ₃	0.730	0.213	0.753	3.42	10.79
+Ag	0.430	0.112	0.442	3.84	10.71

Bulk density and open porosity of the sintered pellets were obtained using Archimedes principle and employing boiling water method. The skins of the sintered pellets were lightly abraded to clear-off the unwanted loose particles pairs to density measurements. Bulk density and open porosity of the samples were measured using Archimedes principle and employing boiling water method. Bulk densities of some of the composite pellets were found nearly 3 g/cc. Attempt should be made to improve the density in order to use the developed powder as cutting tool material.

3.4 Hardness of the developed pellets

Table 3.6 Hardness of the developed pellets

Material	Load (gf)	Time (sec)	(d1)µm	(d2) µm	Hardness (HV)
	200	15	19.5	19.5	975
Al_2O_3	200	15	19	19	1027
	200	15	19	19.5	1001
Al ₂ O ₃ +	200	15	20	19.5	955
Ag	200	15	19.5	20	950

One of the most useful properties of tools is hardness. The hardness of a material is usually defined as the resistance to permanent deformations like abrasion wear, indentation, scratching, grooving etc. The hardness of a material is a direct result of interatomic forces acting on the surface, which again depends upon the strength of chemical bonding, crystal structure and the plastic deformation. The hardness of the present ceramic pellets has been measured in a Vickers Hardness tester. Ceramic cutting tools can be used in metal cutting, and can keep their hardness, strength, abrasion resistance and long performance life under high cutting speed. With these special mechanical properties, ceramic cutting tools will certainly substitute high speed steel cutting tools in many fields.

3.5 Indented micrograph of developed pellets



Fig 3.6 Indented with 200gf load of Sample 3



Fig 3.7 Indented with 200gf load of Sample 4



Fig 3.8 Indented with 200gf load of Sample 5

Figure 3.6, 3.7, 3.8 are the sample indented micrograph of the samle-3, 4, 5 respectively for the measurement of the hardness. The maximum hardness was 1027HV.

4. Conclusions

- Initial powder of α-alumina and silver nitrate can easily be obtained through chemical synthesis
- Alumina-silver composite can be obtained from the initial powder of α-alumina and silver nitrate from conventional sintering root.
- X-ray diffraction analysis indicates presence of only α-alumina and silver in the composite
- Obtained density of alumina-silver composite pellet is quite well
- Varying the process parameter of compaction and sintering, higher density of the composite is expected to be achieved which will result in silver toughened alumina composite that in turn will lead to development of advanced ceramic cutting tool material.

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